

EFFECT OF THE SINTERING ON MICROSTRUCTURE AND SUPERCONDUCTING PROPERTIES OF $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ CERAMICS

Reçu le 05/06/2004 - Accepté le 24/04/2005

Résumé

Les céramiques supraconductrices $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ (Y-123) ont été préparées par la méthode de la réaction à l'état solide. Les propriétés microstructurales et supraconductrices de nos échantillons frittés dans des conditions variées, ont été caractérisées par la diffraction des rayons X (XRD), la microscopie électronique à balayage (MEB), l'analyse thermique (TDA-TGA) et les mesures de résistivité électrique. L'étude montre que la calcination à 800°C pendant 30 heures, du mélange (Y_2O_3 , BaCO_3 CuO), conduit à un début de formation des phases Y-123 et BaCuO_2 . Cependant, dans l'intervalle de température 950-990°C, nous observons alors la cristallisation de Y-123 qui devient la phase prédominante. Pour des températures relativement plus élevées (aux environs de 1100°C) se forme la phase dite "verte" Y_2BaCuO_5 . Enfin, l'orientation préférentielle des grains Y-123, suivant les plans (001), a lieu après frittage à l'air libre dans l'intervalle 950-990°C pendant 30 heures. Ainsi, les échantillons texturés présentent de meilleures propriétés de transport.

Mots clés : Supraconducteur à haute température critique $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$, Diffraction des rayons X (XRD), Microscopie électronique à balayage (MEB), L'analyse thermique (TDA-TGA), Mesures de résistivité électriques.

Abstract

Superconducting $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ (Y-123) ceramics were prepared by the solid-state reaction method. The microstructure and superconducting properties of the sintered samples obtained under various conditions were investigated by means of X-ray diffraction (XRD), scanning electron microscopy (SEM), thermal analysis (TDA-TGA) and electrical resistivity measurements. The study showed that the calcination at 800°C for 30 hours of the Y_2O_3 , BaCO_3 and CuO mixture led to the starting of the Y-123 and BaCuO_2 phases' formation. However, in the temperature range of 950 - 990°C, Y-123 crystallized and became predominate compound. For higher temperatures (near 1100°C), green Y_2BaCuO_5 phase was formed. A (001) preferred orientation of Y-123 grains was occurred after sintering in air at temperatures 950-990°C for 30 hours. The textured samples presented the best transport properties.

Keywords: $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ high-Tc superconductors, X-ray diffraction (XRD), scanning electron microscopy (SEM), thermal analysis (TDA-TGA), Sintering.

M. MAHTALI
A. BOUBEGHAL
A. BOUABELLOU

Laboratoire des Couches Minces et Interfaces,
Université Mentouri
Constantine. Campus Chaab Erassas, Constantine 25000,
Algérie.

ملخص

حضرت الخزفيات الفائقة الناقلة (Y-123) $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ عن طريق التفاعل في الحالة الصلبة. إن الخواص المجهارية والفائقة الناقلة للعينات المليدة في ضروف مختلفة قد شخصت عن طريق التحليل الطيفي للأشعة السينية (XRD) ، المسح المجهري الإلكتروني (MEB) ، التحليل الحراري (TGA) و قيس المقاومة الكهربائية. فمن هذه الدراسة يتبيّن أن الكلسنة في 800°C لمدة 30 ساعة للخلط المسحوق (Y_2O_3 , BaCO_3 CuO) قد تؤدي إلى بداية تكون الطور بين Y-123 و BaCuO_2 . في مجال درجة الحرارة (950-990°C) يلاحظ تبلور Y-123 الذي يصبح يمثل الطور المهيمن. وأما في حالة الحرارات المرتفعة نسبياً (حوالي 1100°C) يتمثل الطور الأخضر Y_2BaCuO_5 . يحدث الاتجاه التناضلي للجيوبات حسب المستويات (001) بعد عملية التلبد في الهواء في مجال درجة الحرارة 950-990°C لمدة 30 ساعة. وهذا قد تتميز العينات ذات الاتجاه المفضل بأحسن خواص النقل.

كلمات مفتاحية : الفائقة ذات درجة الحرارة الحرجة $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ ، الأشعة السينية (XRD) ، المسح المجهري الإلكتروني (MEB) ، التحليل الحراري (TGA)

The synthesis of high-Tc superconductors by a solid-state reaction method is widely performed because of its simple technique [1, 2]. Therefore, the dependence of the structural and superconducting properties on the growth conditions is of great interest [3]. This paper presents the preparation of Y-123 superconductors from a Y_2O_3 , BaCO_3 and CuO mixture under different heat treatment conditions and the characteristics of prepared ceramics.

EXPERIMENTAL DETAILS

Ceramics samples were prepared by the solid state reaction method. Fine powders of Y_2O_3 , BaCO_3 and CuO in appropriate proportions were thoroughly mixed and calcined in alumina crucibles in the atmosphere at 800-1100°C for a period of 16-100 hours with a heating rate of 2°C/min and a slow cooling. The samples were pressed into pellets form and then sintered at 930-1100°C for a period of 30-50 hours. The oxygenation annealing was realized at 500°C for 24 hours. The samples have been characterized by XRD, SEM and DTA-TGA techniques and through their electrical resistivities.

RESULTS AND DISCUSSION

Figure 1 shows the XRD spectra for the samples calcined for a period of 30 hours at different temperatures. At 800°C (Figure 1a), XRD measurements reveal the formation of superconducting $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ phase of which the proportion of its peaks is in the minority, the presence of the intermediate BaCuO_2 phase as well as the starting products.

At 920°C (Figure 1b), the Y-123 phase appears clearly but with peaks of weak intensity, indicating that the superconducting phase is not well crystallized. At 990°C (Figure 1c), the intensities increase considerably, i.e. the crystallization and the stabilization of the Y-123 phase are well improved. The splitting of the peaks in $2\theta = (32.65^\circ, 32.90^\circ)$ is explained by the increase of the oxygen content [4]. At 1100°C (Figure 1d), XRD spectrum shows that the amount of the Y-123 phase decreases enormously and the green Y_2BaCuO_5 phase appears with a majority proportion.

This phase results from the peritectic decomposition of the Y-123 phase in presence of CuO [5]. After sintering of products, in form of pellets, at 930°C, the texture is not observed (Figure 2a). On the contrary, at 960 - 990°C one notes an important texture according to the (001) direction (Figure 2b, c). Besides, an oxygenation annealing carried out at 500°C for 24 hours improves the texture. However, the texture disappears at 1100°C (Figure 2d).

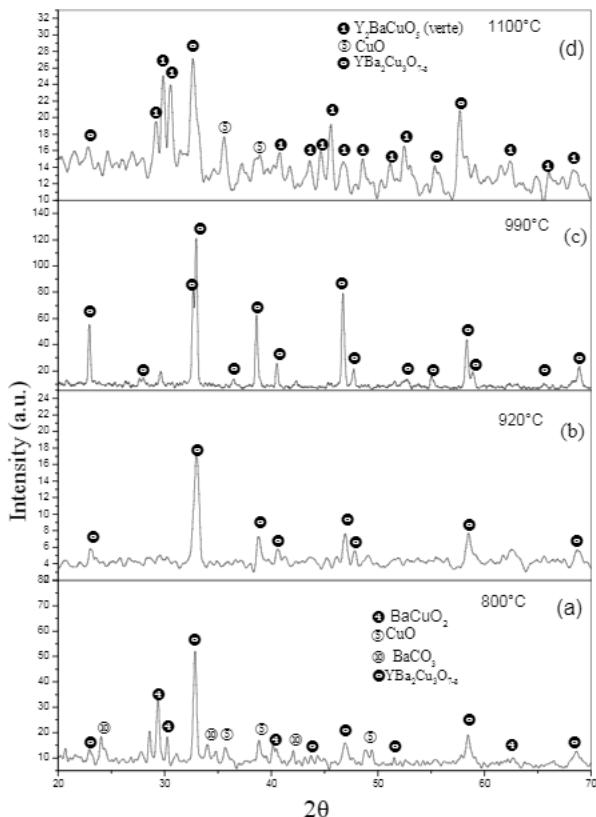


Figure 1: XRD patterns of the powders of the mixture (BaCO_3 , Y_2O_3 , CuO) calcined for 30 hours at 800 (a), 920 (b), 990 (c), 1100°C (d).

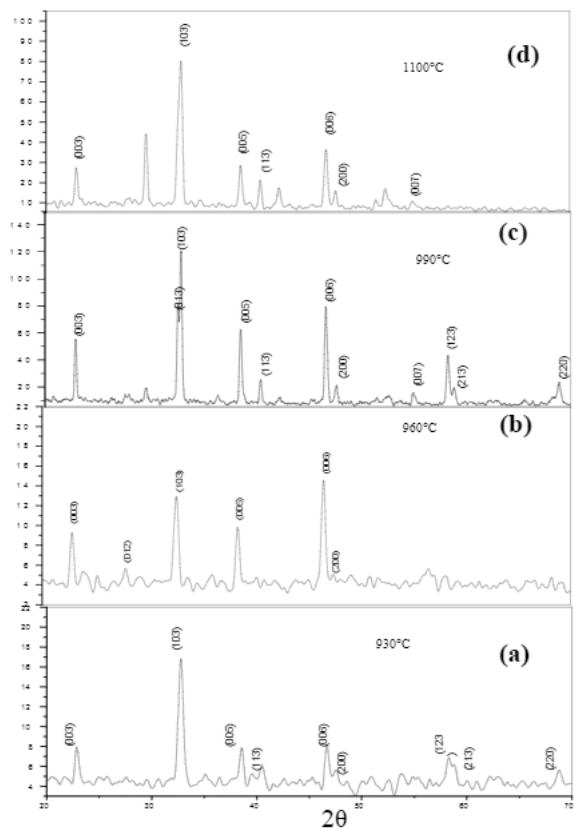


Figure 2: XRD patterns of Y-123 pellets sintered for 30 hours at 930 (a), 960 (b), 990 (c), 1100°C (d).

The Y-123 phase has been also made by use of BaCuO_2 and $\text{Y}_2\text{Cu}_2\text{O}_5$ as precursors, initially obtained by calcination of (BaCO_3 , CuO) and (Y_2O_3 , CuO) mixtures at 990°C for 16 hours respectively (Figure 3 a, b). The pellets of the BaCuO_2 and $\text{Y}_2\text{Cu}_2\text{O}_5$ mixture have been sintered at 990°C for 100 hours (Figure 4). However, the Y-123 phase grains are not textured in this case. The dependence of the electrical resistivity on the temperature of the samples calcined at 940°C for 16 hours, sintered at 950°C for 50 hours (wa) and then oxygenated (sa) is shown in Figure 5. According to the Table, the critical T_c temperature of (sa) sample is higher than that of (wa). The relatively high value of the transition ΔT_c width in (sa) is attributed to the weak-links [6] and the non homogeneity of oxygen vacancies in this sample [7].

Table 1: Critical values obtained from electrical resistivity measurements.

Samples	$T_c^{\text{on}}(\text{K})$	$T_c^{\text{off}}(\text{K})$	$T_c(\text{K})$	$\Delta T_c(\text{K})$
(sa)	97.2	87.3	92.3	9.9
(wa)	92.4	88.1	90.3	4.3

Fig. 6 shows a typical microstructure of samples sintered at 950°C for 50 hours after calcinations (951°C, 30hours). As can be seen, the grains growth, the size of which is ranging from 4 to 45 μm , is not uniform with

prevalence of flattened form and a notable reduction in porosity. The DTA curve of starting powders mixture (Figure 7a) shows the presence of two endothermic peaks at 810 and 895°C attributed to the BaCO_3 decomposition and the beginning of the superconducting Y-123 phase formation respectively.

The TGA analysis of the same mixture (Figure 7b) confirms the BaCO_3 decomposition by a loss of mass at temperatures higher than 810°C. The results of thermal analysis are in good agreement with those of XRD.

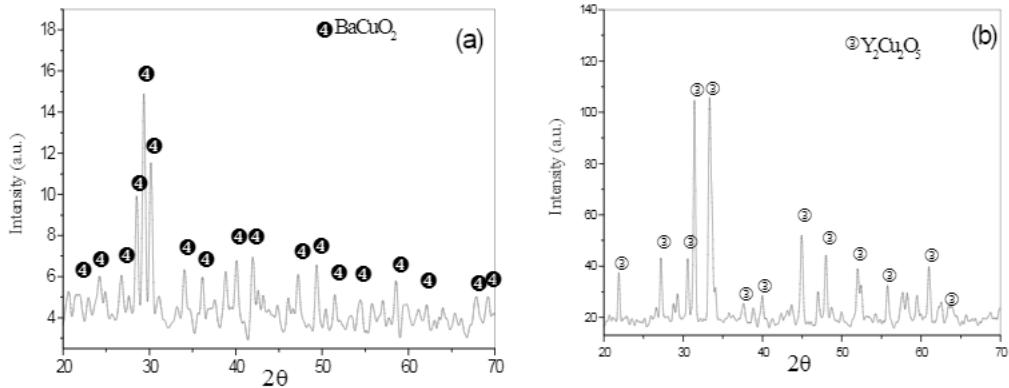


Figure 3: XRD patterns of the powders of mixtures (BaCO_3 , CuO) (a) and (Y_2O_3 , CuO) (b) calcined at 950°C for 16 hours.

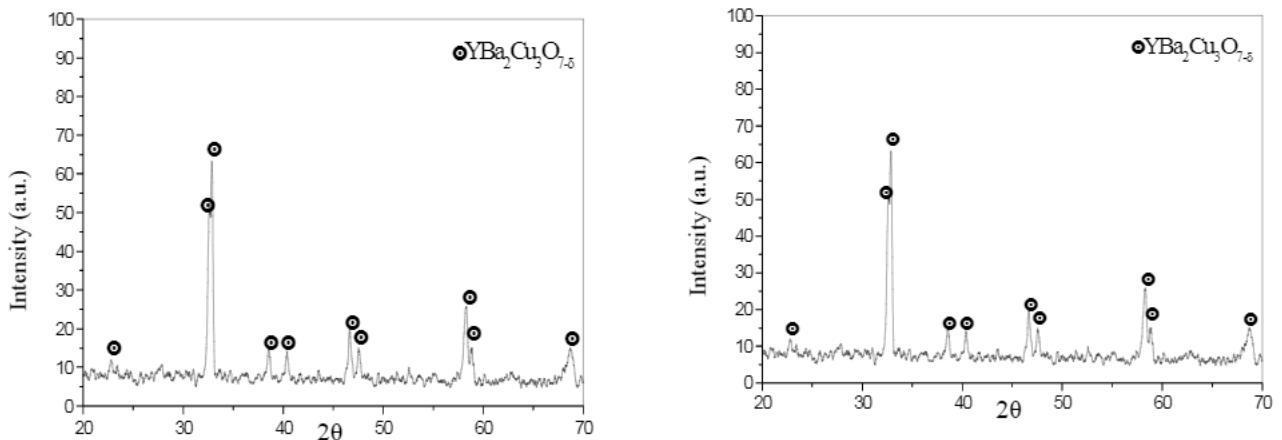


Figure 4: XRD patterns of pellets of the mixture BaCuO_2 and $\text{Y}_2\text{Cu}_2\text{O}_5$ sintered at 990°C for 100 hours.

Figure 5: Temperature dependence of the electrical resistivity of samples calcined at 940°C for 16 hours, sintered at 950°C for 50 hours (wa) and then oxygenated at 500°C for 24 hours (sa).

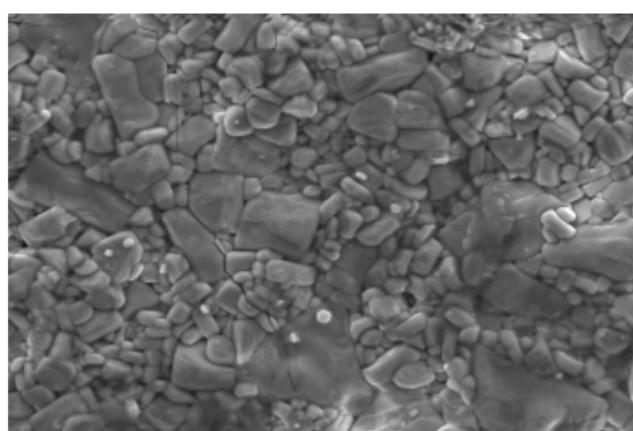


Figure 6 Typical SEM micrograph of pellets sintered at 950°C for 50 hours.

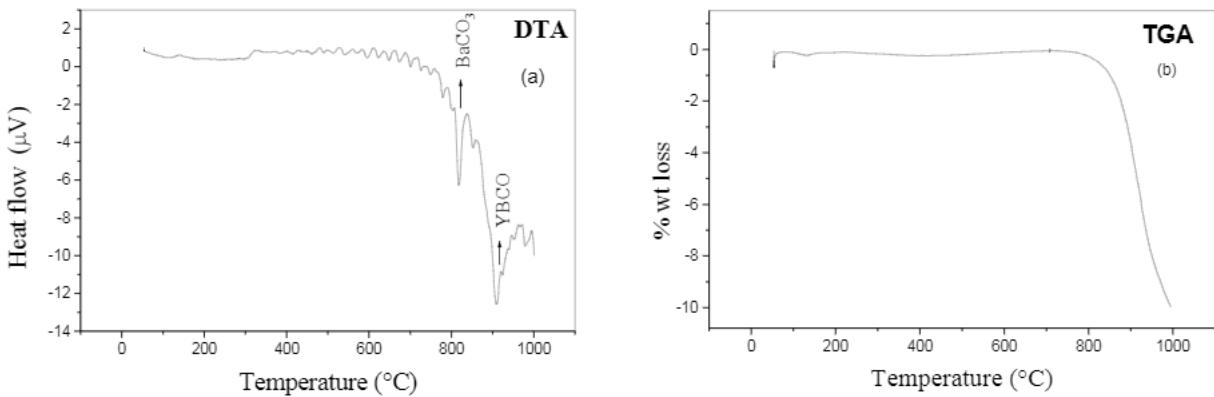


Figure 7: DTA (a) and TGA (b) curves of the powder of the mixture (BaCO_3 , Y_2O_3 , CuO).

CONCLUSION

The sintering in the atmosphere of calcined mixtures (BaCO_3 , Y_2O_3 and CuO) at $950 - 990^{\circ}\text{C}$ for 30 hours or (BaCuO_2 and $\text{Y}_2\text{Cu}_2\text{O}_5$) at 990°C for 100 hours leads to the formation of pure and well crystallized superconducting $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ phase.

The sintering at $950 - 990^{\circ}\text{C}$ provides the texture of Y-123 grains according to the (001) direction. The oxygenation annealing improves the texture.

The electrical resistivity measurements confirm the superconducting character of prepared samples with $T_c=92.3\text{K}$.

REFERENCES

- [1]- J.G. Huber, W.J. Liverman, Y. Xu, and A.R. Moodenbaugh, Phys. Rev. B **41**, 8757 (1990).
- [2]- C.O. Kim, and J.C. Park, J. Mater. Res. **13**, 2067 (1998).
- [3]- M.F. Imayev, D.B. Kazakova, A.N. Gavro, and A.P. Trukhan, Physica C **329**, 75 (2000).
- [4]- A. Nishida, N. Fuketa, K. Furuya, and K. Horia, Jpn. J. Appl. Phys. **33**, 4583 (1994).
- [5]- A.Koblischka et Al, Supercond. Sci. And Technol. **18**, S158-S163 (2005).
- [6]- A.J. Jacobson, J.M. Newsam, D.C. Johnson, D.P. Gorshorn, J.T. Lewandowski, and M.S. Alvarez, Phys. Rev. B **39**, 254 (1989).
- [7]- S. Glenis, G. Choi, C.L. Lin, T. Mihalisin, and X.Q. Wang, J. Appl. Phys. **79**, 5873 (1996).

